

CHAPTER - V

GENERAL DISCUSSION AND CONCLUSION

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This chapter is devoted to the general discussion of the results obtained in the present investigation. Although some interpretations and inferences drawn from specific studies are described at the end of the preceding chapters, they are summarised and further discussed.

5-1 ABSORPTION CHARACTERISTICS:

A) Unfluxed & Fluxed Samples:

The absorption spectrum of undoped and unfluxed CaO found to exhibit a strong absorption band at about 4000 A^{O} alongwith weak bands at about 4200 \mathbf{A}° , 4400 \mathbf{A}° , 4600 \mathbf{A}° , 4800 \mathbf{A}° , 5000 \mathbf{A}° , 5400 \mathbf{A}° and at 5800 \mathbf{A}° with decrease in absorption towards shorter wavelength side (below 4000 A^{O}). Willi Lehmann (1) has reported a strong optical absorption at about 6.5 eV (= 191.2 nm) for purest CaO powder. Moreover, according to Willi Lehmann CaO phosphor cannot be excited optically across the band-gap using 254 nm (= 4.9 eV) radiations. The optical spectra in the present investigation has been studied in the wavelength range 350 nm (= 3.5 eV) to 600 nm (= 2.07 eV), the energy too small to excit the CaO across the band gap. Thus, absorption bands observed for unfluxed and undoped CaO could be attributed to the self activation of CaO, the lattice defects created during synthesis of phosphor at high temperature and/or to the presence of unknown impurities (9). Addition of flux found to suppress the band at about 4000 A° with other bands remaining almost at the same wavelength and this suggests that incorporation of flux does not introduce new absorption centers but probably it causes for neutralization of the absorption centers responsible for 4000 A° band. Incorporation of flux in presence of either Bi³⁺ or Sm³⁺ does cause for appearence and disappearence of some of the bands and to change amounts of absorption either towards shorter wavelength side or longer wavelength side, but observed effect is not systematic and significant. The observed effect might be due to the slight perturbations in peak positions and amounts of absorption within the limits of statistical variations.

B) CaO:Bi;Sm Phosphors (Fluxed):

Undoped CaO found to exhibit absorption bands at about 4200 \mathbf{A}° , 4400 \mathbf{A}° , 4600 \mathbf{A}° , 4800 \mathbf{A}° , 5000 \mathbf{A}° , 5400 \mathbf{A}° and at 5800 \mathbf{A}° . Addition of activator Bi³⁺ does give rise to new absorption bands but the observed bands appear and disappear with an increase in Bi³⁺ concentration in an unsystamatic manner. Moreover, higher concentration of Bi³⁺ found to increase the intensity of absorption bands in 3500 \mathbf{A}° to 4500 \mathbf{A}° wavelength range with the enhancement of absorption towards the longer wavelength side.

Incorporation and changing concentration of Sm³⁺ found to cause for appearence and disappearence of some of

the bands and also to change the band intensities, but the effect is unsystematic and small. Similar are the results obtained with addition of either Bi^{3+} or Sm^{3+} in the presence of each other.

Thus, it may be concluded that whatever effect has been observed with addition of activators Bi³⁺ and Sm³⁺ in peak positions, peak intensities, amounts of absorption either towards shorter wavelength side or longer wavelength side and also about the appearence and disappearence of some of the bands may be attributed to the slight perturbations in the peak positions and amounts of absorption within the limits of statistical variations, caused by activators.

5-2 ORIGIN OF ABSORPTION BANDS & NATURE OF ABSORPTION CENTRES:

A comparision of absorption spectral results obtained for unfluxed and undoped CaO, fluxed CaO and for fluxed CaO:Bi:Sm samples reveal that whatever absorption bands observed for undoped and unfluxed CaO are seen to be almost for all other samples studied except the slight perturbation in peak positions, amounts of absorption and appearence and disappearence of some of the bands within the limits of statistical variations. Thus the observed absorptions bands is probably the property of defects present in host lattice and not that of flux or activator. The defects which are likely to be present in CaO lattice are Ca²⁺ and O²⁻ vacancies or intersitials. To elucidate the nature of these defects EPR measurements have been carried out but as M_n^{2+} is found to be present invariably in all samples as a trace impurity, our efforts could not meet with success. The defects present in CaO lattice might be created during the synthesis of phosphor at high temperature and/or due to the presence of unknown trace impurity in ingradients.

5-3 <u>EMISSION CHARACTERISTICS</u> - Origin of Spectral Bands and Nature of Luminescence Centres:

A) Undoped & Unfluxed CaO:

It has been found that unfluxed and undoped CaO exhibits bands at about 4250 A° and at 5890 A° . The band at about 4450 A° is weaker while 5890 A° band is strong. The 5890 A° band is invariably observed for all samples and it has been attributed to M_n^{2+} impurity. The detailed discussion is left for the next section. The band at about 4250 A° might be due to self activation of CaO.

B) CaO (With Flux):

Addition of flux causes to give rise to additional band at about 5250 \mathbf{A}° with other bands 4500 \mathbf{A}° and 5890 \mathbf{A}° remaining almost at same positions. The fluxes added are Na₂S₂O₃, Na₂SO₄ and Naf. We believe that out of Na¹⁺, SO₂²⁻ and F¹⁻ ions probably either Na¹⁺ or F¹⁻ ions are responsible for the 5250 \mathbf{A}° band. However, as reported by Shalgaonkar (2) Na^{1+} has closed shell structure and does not have any energy levels even with 10 eV of the ground state and as such the observed band will not be a characterstic property of Na^{1+} ion. Hence here we attribute the 4450 a° band to the F^{1-} ion.

C) CaO:Bi (With Flux):

The emission spectra of CaO:Bi (fluxed) phosphors found to exhibit bands peaking at about 4250 A° , 4500 A° , 5250 A° and 5890 A° . The peak intensities of 5250 A° and 5890 A° are higher than the other ones.

The electron configuration of Bi atom is 15^2 , 25^2 , $2p^6$, 35^2 , $3p^6$, $3d^{10}$, $4f^{14}$, $5s^2$, $5p^6$, $5d^{10}$, $6s^2$, $6p^3$ and the ground state of free ion is 15° .Bi³⁺ enters in trivalent states loosing $6p^3$ electrons. The first excited state is due to 65, 6P electron configuration and that is P state. According to L-S coupling scheme and Hunds rule the P state is further split into 3P_0 , 3P_1 , 3P_2 and 1_{P_1} (2). Transitions between these levels take place in accordance with the selection rule $\Delta J = \pm 1$. Consequently, the possible allowed transitions are ${}^1P_1 \longrightarrow {}^1S_0$ and ${}^3P_1 \longrightarrow {}^1S_0$. The energy level diagram alongwith allowed and forbidden transition is shown in fig.5.1. The allowed transitions i.e. ${}^1P_1 \longrightarrow {}^1S_{0A}$ should give emission at about 4470 A° and $5120 A^\circ$ respectively (2,3). In fact such emissions have been observed at about 4500 \mathbf{A}° and 5250 \mathbf{A}° for samples containing Bi³⁺ but these are found to overlap on the emission peaks corresponding to the fluxed CaO. The emission band found at about 4250 \mathbf{A}° may be due to a luminescence centre formed by Bi³⁺ impurity in co-operation with the other local defects.

D) CaO: Sm Phosphors (With Flux):

Incorporation and changing concentration of Sm^{3+} found to affect sensitively emission spectra. A new emission peak at about 4250 A° has been observed and emission intensity continuously go on increasing towards the orange-red region with disappearence of peak at about 5250 A° . Moreover, addition of Sm^{3+} found to cause decrease in intensity of a 4500 A° peak.

Sm is one of the Lanthanide series of rare-earth elements having electron configuration (Xe) $4f^6$, $5d^\circ$, $6s^2$ and is characterised by unfilled 4f shall electrons shielded by $5d^\circ$, $6s^2$ electrons. The energy level diagram of Sm³⁺ ions in LaCl₃ as given by G.F.Imbusch (4) is shown in fig.5.2. The transitions shown become allowed due to mixing a state when Sm³⁺ is incorporated in lattice. In fig.5.3 is shown the possible transitions of Sm³⁺ alongwith the emissions while energy term intervals between the six levels of the ground state are summarised in table 5.1. These results are reported by S.Makishima et.al (5) for Sm³⁺ in BaTiO₃ matrix. From these results it is clear that when Sm³⁺ is incorporated in any lattice it emit in orangered region. Cur results also show the similar emissions. However we could not attribute a particular emission to a specific transition because of limitations put by experimental procedure followed in recording the emission spectra.

E) Origin of 5890 A^O Band:

The intense emission band around 5890 A^{O} is found invariably for all samples. The same band was also observed by many workers in an undoped CaS and ZnS (2,6,7,8,9). At present there is great contraversy regarding the origin of this band. Abhyankar (10) has reported this band is CaS:Ce phosphors to be due to Na¹⁺ ions introduced through flux Na_2SO_4 . Our samples also contain Na_2SO_4 . $Na_2S_2O_3$ and NaF as flux and hence it is tempting to attribute this band to Na¹⁺ ions. But, as stated earlier Shalgaonkar has reported that Na¹⁺ ion has closed shell structure and does not have any energy levels even within loeV of the ground state and as such the observed band will not be the characteristic property of Na¹⁺ ions. On the other hand Lalita et.al (11) reported that the deliberate additon of M_n^{2+} to CaS substitutes for Ca²⁺ to M_n^{2+} and gives rise to an emission band at about 5890 A°. Moreover, Lehmann (12) and Hadgal (9) have reported that an undoped CaS and ZnS contains M_n^{2*} as trace impurity and is responsible for emission at 5890 A°. Our EPR spectra of doped and undoped

CaO phosphors clearly show the presence of M_n^{2+} suggesting thereby the band at about 5890 A° is a characteristic of M_n^{2+} impurity which might be present as a trace impurity in starting materials. This view is supported by the results of Shankar et.al (6) on CaS phosphors.

F) CaO:Bi:Sm Phosphors (With Flux):

When either Bi^{3+} or Sm^{3+} is added and varied in presence of each other both of them found to affect the emission spectra sensitively with giving their own characteristic emission.

5-4 MECHANISM OF ENERGY TRANSFER:

In luminescence transfer of energy from absorption centre to radiative center may take place either of one or more processes of the followings:-

- i) Cascade mechanism,
- ii) Resonance transfer,
- iii) Exciton migration, and;
- iv) Movement of charge carriers.

An insight into this can be had from absorption and emission spectra.

The measurements of emission spectra reveal that when activator Sm^{3+} is added to the host lattice, it absorbs the emission at about 5250 A° and emit it as its own characteristic emission. These results thus implies that

the cascade mechanism of energy transfer is likely to be operative in CaO:Bi:Sm phosphors. Moreover, the transfer of energy with movement of charge carriers is also an expected process since, motion of electrons is involved during the excitation and emission processes. Here it should be mentioned that we could not draw any conclusion about the operativeness of resonance transfer and exciton migration mechanisms for the studies made in the present investigation.

5-5 X-RAY DIFFRACTION STUDIES:

Lehmann (1) has reported that CaO crystallizes in the same cubic NaCl lattice as CaS. The x-ray diffraction pattern of CaO prepared by thermal reduction method in the present investigation shows a face centered cubic structure with lattice constant 'a' equal to 8.71 A° . The analysis of XRD pattern obtained for fluxed CaO:Bi and CaO:Sm samples found to show the face centered cubic structure with shifting of peak positions towards the higher angle side and decrease in peak heights. The shifting of diffraction lines towards the higher angle side indicates the decrease of lattice constant 'a' with the addition of activators, while decrease in peak heights suggest the reduction in number of atoms in a plane. From the last result, the inference to be drawn is that Bi³⁺ and Sm³⁺ in **CaO** lattice probably take the Ca^{2+} sites. This is because Sm^{3+} and Bi^{3+} are trivalent and Ca²⁺ is divalent and as such the introduction of either

Bi³⁺ or Sm³⁺ at lattice sites should result in reduction of number of atoms in a plane to maintain the neutrality of the crystal. Our results agree with the investigation of Willi Lehmann as CaO phosphors activated with Bi³⁺ and Sm³⁺ (1). The above conclusion is also supported by the ionic raddi of Ca²⁺, Bi³⁺ and Sm³⁺. The ionic raddii of Ca²⁺, Bi³⁺ and Sm³⁺ are 0.99 A⁰, 0.99 A⁰ and 0.98 A⁰ respectively. The difference in ionic raddi is within 15 % and as such Bi³⁺ and Sm³⁺ can take easily the Ca²⁺ position at the lattice site.

5-6 ELECTRON PARAMAGNETIC RESONANCE STUDIES:

The EPR measurement of prepared samples are carried out with a view to gain the information about the defect structure responsible for luminescence properties. However, in all samples studied the characteristic Mn^{55} spectrum is observed. We have not added Mn^{24} as an activator but it might be present as a trace impurity and might have introduced through the starting material $CaCO_3:5H_2O$. Such EPR spectrum of M_n^{24} as a trace impurity was observed by Shankar et.al (6) in CaS phosphors and by Hadgal (9) in (Zn,Ca)S:Bi:Dy phosphors.

The EPR measurements of samples containing Bi^{3+} and Sm^{3+} were carried out but no signal corresponding to Bi^{3+} and/or Sm^{3+} is observed. This is since Mn^{2+} is very sensitive to EPR measurements and other impurities in its presence cannot be detected. Same reasoning applies to the self activated and other defect centers present in CaO lattice.

Addition and change in percentage of Sm^{3+} found to change the crystal field surrounding the Mn²⁺ ion and to suppress the allowed and forbidden transitions except the transition corresponding to $M = +\frac{1}{2} - - -\frac{1}{2}$. Thus indicating thereby probability of formation of the bonding of Sm atoms with the host lattice (9). No such results have been observed for Bi³⁺. The spin Hamiltonian parameter of Mn has been calculated and it is found to be in good agreement with the results of other workers (9,13,14). The calculated value of 'g' varies between 1.99921 to 2.00277 but the observed variation is unsystematic and insignificant.

5-7 CONCLUSIONS:

The principal findings of the present investigation may be summarised as follows -

1) The observed absorption bands is likely to be the property of host lattice defects. These defects are probably the Ca^{2+} and O^{2-} vacancies or interstitials created during the synthesis of phosphor at high temperature.

2) Whatever effect has been observed with addition of flux and activators (Bi^{3+} , Sm^{3+}) in peak positions, amounts of absorption and about the appearence and

disappearence of some of the bands may be due to the slight perturbation in the peak positions and amounts of absorption within the limits of statistical variation, caused by the flux and activators.

3) Addition of flux introduces a luminescence centre into the host lattice corresponding to a emission at about 5250 A° . The defect could be attributed to the presence of F^{1-} ions in host lattice.

4) Spectral distribution of emitted radiation is significantly affected by the addition and change in percentage of Bi^{3+} and Sm^{3+} .

5) Addition of Bi³⁺ introduces luminescence centres which give their own characteristic emission.
6) Addition of Sm³⁺ introduces luminescence centres which give their own characteristic emission.
7) Bi³⁺ and Sm³⁺ probably take the lattice site positions by replacing Ca²⁺ ions.
8) The band at about 5890 A⁰ is due to the Mn²⁺

trace impurity.

9) The prepared CaO:Bi:Sm phosphors crystallize in face centered cubic lattice with lattice constant $a = 8.71 A^{\circ}$.

10) EPR measurements reveal the presence of Mn^{2+} as trace impurity.

11) There is no significant change in 'g' values.

12) A change in crystal field symmetry around the Mn^{2+} ions with change in percentage of Sm^{3+} points the formation of bendings of Sm^{3+} with the lattice. 13) The processes which are likely to be involved in transfer of energy from absorption centers to the emission centers are cascade mechanism and movement of charge carriers. Moreover, presence of resonance transfer and exciton migration are also possible mechanisms but it is not possible to draw any such conclusion from the present experimental investigation. 14) CaO is vergatile as a phosphor host.

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$^{6}H_{7/2} - ^{6}H_{9/2}$	1431	9.1	1160	8.2	8.5
${}^{6}H_{9/2} - {}^{6}H_{11/2}$	1531	9.8	1293	9.1	9.5
$^{6}H_{11/2} - ^{6}H_{13/2}$	1	i	1332	9.5	10.1
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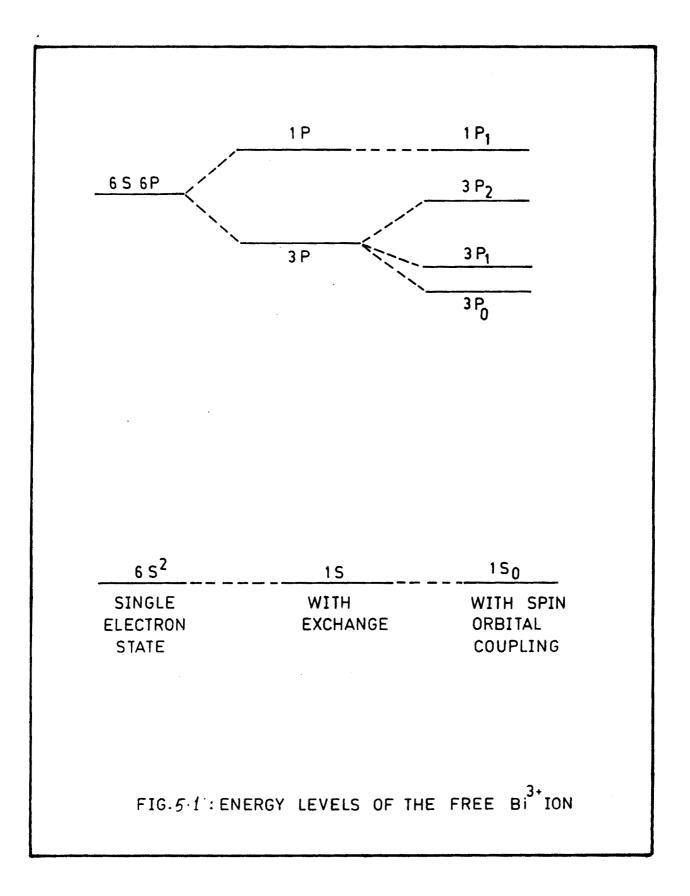
Energy term intervals between the six levels of the ground state of Sm^{3+} .

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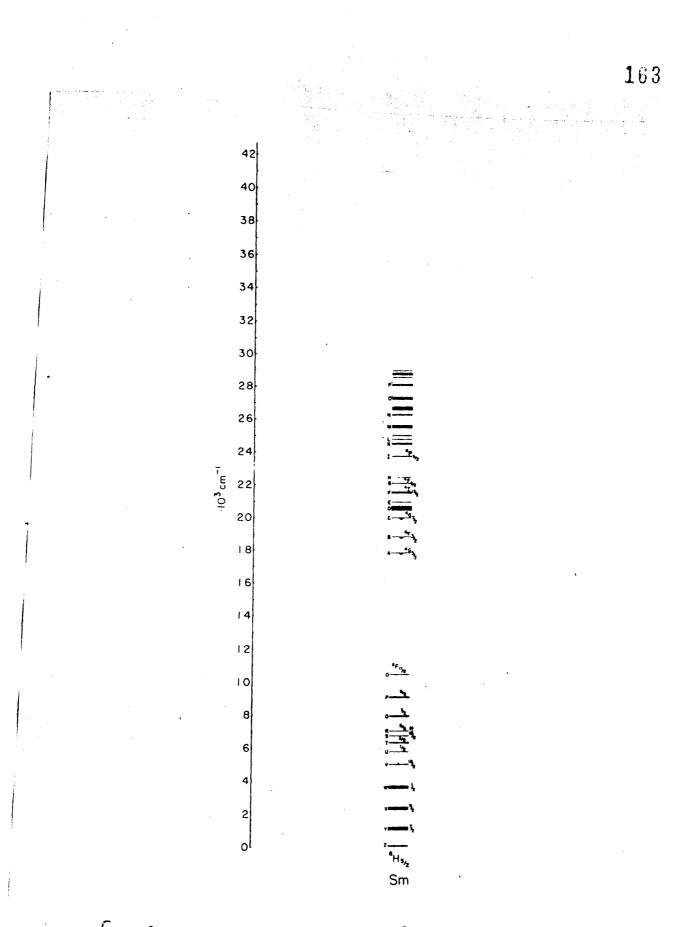
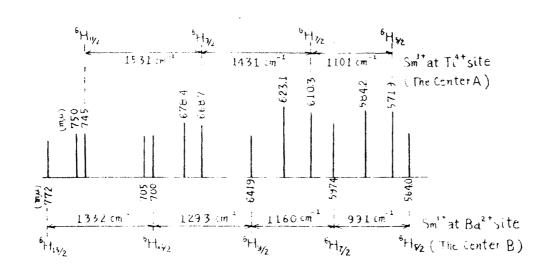


Fig. 5.2: Energy level diagram of sm ions in Lacks.



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Fig.5.3 Diagram of the emission spectrum of Sm^{3+} in a BaTiO, metrix at -196 C.